

## 1. Introduction

Starch and cellulose based materials derived from naturally occur biopolymers are the major pharmaceutical excipients utilized in drug delivery dosage forms. These polymers always interact with water due to their hydrophilicity exhibiting some properties that may critically affect the dosage form performance. For example: In controlled release devices, water diffusion through a polymeric hydrogel layer has been considered as one of the major factors determining drug release rate (Rajabi-Siahboomi, et al., 1996). With liquid water in excess, these hydrophilic polymers could form hydrogels i.e., the three-dimensional arrangement possessing the ability to retain a significant fraction of water without complete dissolving. A hydrogel might form relatively stable space lattice or network pores fulfilled with a considerable amount of water. The interfacial tension related to surface of curvature of water within pores could develop and affect the phase transition of the water. Thus this phase transition of water confinement could somehow characterize the pores where it occupies. A number of authors, for examples: Yamamoto et al. (2005), Faroongsarn & Peck (2003), Hay & Laity (2000), and Ishikiriyama & Todoki (1995) examined the pore sizes and distributions of various porous materials assuming that water is mostly held within pores, with melting temperature being depressed by Gibbs-Thomson effect. However, the depression of melting temperature is not only attributed by water confinement in porosity but the water-polymer interaction. Rault et al (1994) reported that the melting depression and the concentration of unfrozen water varied with the water concentration with similar orders of magnitude for polymer-water systems and simple binary mixtures, presenting the same type of interaction, from which confinement effects are absent. They concluded that the melting depression is due not to water confinement in polymer network porosity but rather to water-polymer interactions. The evidence was later confirmed by the work of Okoroafor et al (1998).

In general, interactions between macromolecules fall into four categories: ionic, hydrophobic, van der Waals and hydrogen bonding (Ilmain, et al., 1991). But for a polymer-water mixture, the interaction is always in the range of hydrogen bonding. It has been proposed (Ping, et al., 2001; Zografi & Kontny, 1986; and Higuchi & Iijima, 1985) that water in hydrophilic polymer matrices presents in three distinct fractions: (i) non-freezable bound water, (ii) freezable bound water, and (iii) free or bulk water. Upon cooling, water begins to crystallize only when its content is above a characteristic threshold. This fraction of water has been called freezable bound water (fraction (ii)) because it exhibits a melting point lower than zero °C which is distinguished from bulk water and it should correspond to the depression phenomenon described above. In the lower-than-threshold level, i.e., the water of fraction (i), the molecules of water interact with polar functional groups such as carboxyl groups on polymer chains. The interaction would be well-oriented hydrogen bonding which is locally favorable configuration that being strong enough to prevent water to form ice crystals (Ping, et al., 2001). The differential scanning calorimetric (DSC) study can reveal the freezable water fractions, for example: Nakamura et al. (1981) reported two DSC peaks of crystallization of absorbed water on celluloses including a broad peak observed at ~230-250 K and a sharp one at ~255 K. Should the melting depression of water of fraction (ii) be due to polymer-water interaction, the corresponding DSC peak then could describe the thermodynamics of a polymer-water system. Many techniques are available for the experimental determination of the interaction parameter between solvent molecules and the polymeric chain segment. However, the methods were usually based on volumetric determinations (Mantovani, et al., 2000). The melting/freezing depression determined by DSC could also exhibit the great potential to characterize that interaction. The aim of the study is to examine

the thermal behavior of water melting depression due to its interaction with the selected starch and cellulose based polymers commonly used in drug delivery formulations by mean of DSC technique.

## 2. The thermodynamic relations for a polymer solution

A general thermodynamic theory of polymer solution based on mixing according to liquid lattice theory has been presented by Flory (1971). For polymeric hydrogels employed in the present study, the chemical potential of water ( $\mu_1$ ) in a water-polymer system includes not only Flory's mixing with swollen gel but the Donnan equilibrium for polyelectrolytes that yields the following relationship (Flory, 1971; Okoroafor, et al., 1998; Mantovani, et al., 2000; Ozmen & Okay, 2005; and Borchard, et al., 2005):

$$(1) \quad \mu_1^{gel} - \mu_1^0 = RT[\ln(1 - \varphi_2) + \varphi_2 + \chi_1 \varphi_2^2 + V_1(\frac{v_e}{V_0})(\varphi_2^{1/3} - \frac{\varphi_2}{2}) - f \cdot \varphi_2]$$

Where,  $\varphi_2$ ,  $\chi_1$ ,  $v_e$ , and  $f$  are volume fraction of polymer in gel, the Flory's polymer-water interaction parameter, the effective crosslink density of the network, and the fraction of charged units in the hydrogel network, respectively.  $V_1$  and  $V_0$  are molar volume of water and the volume of relaxed hydrogel network.  $R$  and  $T$  are gas constant and absolute temperature.  $\mu_1^0$  is the chemical potential of pure liquid water. And,  $\mu_1^{gel}$  is the chemical potential of water in hydrogel. The first three terms in the right hand side of equation 1 represent the chemical potential of general polymer-water mixture. The fourth term is the chemical potential due to reaction of the network crosslink structure (Flory, 1971), whereas the last term is that from Donnan equilibrium theory (Mantovani et al. 2000; and Ozmen & Okay, 2005).

It is further assumed that frozen water is in equilibrium with the unfrozen water in gel phase during the DSC operation, i.e., the chemical potential of freezing

ice ( $\mu_1^{ice}$ ) and of water in hydrogel ( $\mu_1^{gel}$ ) must be equal. And when a mixture freezes, one of the colligative properties known as freezing point depression holds. The change of chemical potential can be written as (Ozmen & Okay, 2005):

$$(2) \quad \mu_1^{ice} - \mu_1^0 = \Delta H_m \left( \frac{T}{T_0} - 1 \right)$$

Where  $\Delta H_m$ , and  $T_0$  are molar enthalpy of crystallization (or melting), and melting temperature of pure water, respectively. Since the left hand side of equation 1 and 2 are equal, the arrangement of these two equations yields:

$$(3) \quad \frac{1}{T} = \frac{1}{T_0} - \frac{R}{\Delta H_m} \cdot [\ln(1 - \varphi_2) + \varphi_2 + \chi_1 \varphi_2^2 + V_1 \left( \frac{V_e}{V_0} \right) (\varphi_2^{1/3} - \frac{\varphi_2}{2}) - f \cdot \varphi_2]$$

This equation should be applicable to the water of fraction (ii) where the ice-liquid water transition temperature was depressed. And, assuming the involved parameters are constant over the transition temperature, the parameters such as  $\chi_1$  could be obtained by non-linear regression of  $\frac{1}{T}$  as a function of  $\varphi_2$  according to the model described by equation 3.

### 3. Materials and Method

#### 3.1 Materials

The variety in nature of starch and cellulose based polymers including pre-gelatinized potato starch (PS: Starch<sup>®</sup> 1500, Colorcon, Inc., PA, USA), sodium alginate (SA: Wendt-Chemie, Hamburg, Germany), sodium starch glycolate (SSG: Explotab<sup>®</sup>, JRS Pharma, Rosenberg, Germany), hydroxypropyl methyl cellulose (HPMC: Colorcon, Inc., PA, USA), Sodium carboxymethyl cellulose (SCMC: Wendt-Chemie, Hamburg, Germany) and croscarmellose sodium (CCS: Ac-di-sol<sup>®</sup>, FMC Corp. PA, USA) were employed. SA, PS, and SSG were charged-linear, branch

and linear, and charged-crosslinked polysaccharides, respectively. HPMC, SCMC, and CCS were linear, charged-linear, and charged-crosslinked celluloses, respectively.

### 3.2 *Sub-ambient differential scanning calorimetric study*

The Perkin-Elmer differential scanning calorimeter (DSC7 with TAC7/DX Thermal analysis controller, Perkin-Elmer Corp., Norwalk, CT, USA) equipped with liquid nitrogen bath set as a cooling accessory was employed. Calibrations with Indium and cyclohexane were carried out for every time which the DSC operation started to ensure the accuracy/precision of the obtained heat of transitions and the corresponding temperatures. An accurately weighed (5-15 mg) sample was placed in tightly sealed aluminum pan (Perkin-Elmer Corp., Norwalk, CT, USA). The samples were subjected to run against an empty pan as a reference. With loading temperature of 25 °C, the analysis program includes 1) cooling from 25 °C to -150 °C at 5.00 °C/min rate, 2) isothermal run at -150 °C for 1 min, and 3) heating from -150 °C to 25 °C at the same rate as cooling step. The distilled water was run to validate the temperature and heat of water crystallization/melting. All of DSC thermograms (cooling or heating traces) were analyzed using Pyris<sup>®</sup> software (Perkin-Elmer Perkin-Elmer Corp., Norwalk, CT, USA).

The samples were pre-treated with ambient humidity prior to DSC analyses. The ~5 g-samples were equilibrated with 85, 96, and 100% relative humidity (RH) at 30.0±0.2 °C for 10 days. The samples were also fully hydrated by liquid water in excess at the same temperature as those pre-treated with ambient humidity as follows: A 3- to 8-gram sample (equivalent to approximately 10-ml bulk volume) was thoroughly mixed with liquid water to 100 ml in volume. The mixtures were allowed to be still for 1 day. Hydrogels or sediments depending to the nature of water-polymer mixtures were subjected to sub-ambient differential scanning calorimetric study

described above. The total water ( $W_T$ ) contents of hydrogel/sediment samples were determined using a moisture balance (Mettler® LP16 & PM300, Mettler-Toledo, Inc., Hightstown, NJ, USA) with heating temperature of 100 °C.

### 3.3 The determination of non-freezable water

The water of fraction (i) was calculated by subtracting the total water content ( $W_T$ ) by the water content calculated from the amount of heat corresponded to DSC melting traces in sub-ambient temperatures assuming that the area of melting peak of pure water corresponds to the melting enthalpy. So, the heat was converted to the amount of water since it was directly proportional to enthalpy of melting obtained from DSC tracing of distilled water.

### 3.4 The determination of polymer volume fraction in liquid water

The fully hydrated polymer volume fraction ( $\phi_{2V}$ ) was obtained from particle size determination in non-swelling and swelling states, as analogous to what was done previously (Mantovani et al., 2000). The size and distribution of each of the polymeric powders were measured by dynamic laser light scattering technique (Mastersizer®/E, Malvern Instruments Ltd., Worcestershire, UK). Alcohol and water were used as non-swelling and swelling media, respectively.  $\phi_{2V}$  was obtained by comparing mean volume diameters according to the equation of  $\phi_{2V} = [\overline{d_{al}}/\overline{d_w}]^3$ , where  $\overline{d_{al}}$  and  $\overline{d_w}$  are geometric mean volume diameters of a powdered polymer in alcohol and in water, respectively.

To quantify the polymer volume fraction during ice-liquid phase transition of water denoted by  $\phi_2$ , it was assumed that only pure water freezes when cooled to freezing point.  $\phi_2$  is thus directly proportional to the cumulative partial area under the DSC peak at corresponding  $T$ , i.e.,  $\phi_2 = \phi_2^{(i)} - \Lambda \frac{AT}{P}$ . Where,  $A$ ,  $T$ ,  $P$ ,  $\phi_2^{(i)}$ , and  $\Lambda$  are

the area under the peak at temperature  $T$ , the total area under the peak, the polymer volume fraction with water of fraction (i), and the linear coefficient that makes  $\phi_2$  equals  $\phi_{2V}$  determined by light scattering technique, in which  $A_T$  equals  $P$ , respectively.  $\phi_2^{(i)}$  was approximated from mole fraction of water of fraction (i) ( $x_1^{(i)}$ ) calculated based on the water content of non-freezable water previously described, i.e.,  $\phi_2^{(i)} \approx (1 - x_1^{(i)})$ . The  $\phi_2$  and its corresponding  $T$  were non-linearly fitted into Flory's model using the commercial software (SigmaPlot<sup>®</sup> 2000, SPSS, Inc.).

#### 4. Results and Discussion

##### 4.1 *In situ* water crystallization: the validation of DSC measurement

The cooling and heating traces revealing water crystallization and melting, respectively, are in Figure 1. There was an exothermic peak of water crystallization (I in Figure 1) occurred at a temperature far below zero °C. Endothermic melting peak (II in Figure 1), on the other hand, started at a normal melting temperature. This inconsistency between freezing and melting curves is commonly observed in fairly slow rate of scanning (1-10 °C/min). It is because the crystallization difficulty causes an exotherm to appear at a temperature lower than normal. It seems that the melting trace could approach an equilibrium ice-water transition better than cooling counterpart as the tracing was close to 0 °C. Table 1 shows the detailed information of water melting (II in Figure 1) compared with the reference (Dean, 1985).

As seen in Table 1, both onset and heat of melting for pure water agree with the values taken from the reference. Very low deviations, i.e., 0.37% and 1.18% deviate from reference values for onset and heat of melting, respectively, are observed. It has been stated that in typical DSC measurement, the mean error at heating/cooling rate of 1-10 °C/min should not exceed 2.5% (Borchard, et al., 2005).

Thus, the method and its conditions could be used to investigate water crystallization/melting with acceptable precision and accuracy.

#### *4.2 DSC water tracings in the selected hydrophilic polymers and the nature of ice-liquid water transition*

Figure 2 illustrates the tracings of water that could be found in SA, SSG, SCMC, and CCS equilibrated with ambient humidity (85-100% RH). For simplicity, the only tracings of SA-100%RH system are showed. As seen in Figure 2, the freezable water in current study is consistent with previous report (Nakamura et al., 1981). It is then subjected into 2 fractions, i.e., water of fraction (ii) labeled as (I) where freezing/melting happen at a temperature below zero, and that of fraction (iii) labeled as (II) where its transitions are closed to normal melting point. Figure 3 illustrates the DSC freezing traces of CCS with various aqueous level environments including that with liquid water in excess. It is noted that other polymers in this study showed similar patterns. However, the water transition tracings were absent in the cases of PS and HPMC in ambient humidity but fully hydrated samples. PS and HPMC are non-ionic polymers exhibiting less hygroscopic than others. It may be because ionic species and salts could attribute to hydration on polymer they present with and might allow amount of water uptake greater than threshold of non-frozen water to show the DSC tracings of water of fraction (ii) and (iii) in cases of SA, SSG, SCMC, and CCS.

Should the porosity formed by 3-dimentional polymer network govern the freezing/melting point depression, the depressed temperature in various moisture environments of the same polymer which would form similar pore structures might be invariant. Furthermore, if the pores collapse during ice formation, the transition of water of fraction (ii) might be either near or far from that of water of fraction (iii) dependent on the new size of the pores that water occupies after collapsing. As



obviously showed in Figures 3, there are not the cases in the present study. It is observed that the phase transition of water of fraction (ii) always exhibits a pattern as a polymer solution, i.e., the more concentration level of water; the more freezing temperature is depressed. Thus rather than porosity confinement, the freezing temperature may be depressed in accord with polymer-water interaction.

Figure 4 illustrates the endothermic melting traces of SSG with variety of humidity as well as fully hydrated sample. Like freezing exotherms, the melting endotherms of various level of water with polymer samples under study were also in similar patterns. It is observed that the melting of freezable bound water shifts toward the melting of free water. i.e., the two singlet peaks turn to a doublet with increase in water content which is similar to the previous study (Borchard et al., 2005). It may be because water of fraction (ii), during increasing temperature, becoming liquid phase migrates from the vicinity of polymer interaction sites within gel due to hydrogen bonding among water molecules to be in equilibrium again with free water that melt later at a normal melting temperature.

#### 4.3 *Non-freezable bound water*

An attempt at the determination of water of fraction (i) for each of polymer-water systems was made and tabulated in Table 2. The materials under study exhibit the non-freezable water contents of between 9.67% and 26.63% whereas it was previously reported that starches and celluloses exhibited non-freezable bound water contents of 28% (Zhong & Sun, 2005) and 22-25% (Luukkonen, et al., 2001), respectively. McCrystal, et al (1997) estimated the number of moles of non-freezing water per a polymer repeating unit for HPMC gel as approximately 3.8 moles that is corresponding to approximately 10-20% water content dependent on degrees of substitution, while the current study on HPMC is within the range (13.21%, Table 2).

On the other hand, the cross-linked chemically modified starch and cellulose that are more hygroscopic (SSG and CCS) illustrate low level of non-freezable bound water (Table 2). It might be because these materials present more number of ice nuclei, during freezing, that draw more water molecules due to hydrogen bonding to the ice clusters as a process of lowering surface free energy. As a result more portion of freezable water may be detected.

#### *4.4 The volume fraction of polymeric hydrogels vs. melting depression: non-linear fitting to the Flory's model*

The volume fractions in liquid water ( $\phi_2 v$ 's) of fully hydrated polymers under study are tabulated in Table 2. It is noted that  $\phi_2 v$ 's of SA and HPMC have been taken from the reference (Mutalik, et al., 2006) since the polymers dissolved in water and alcohol, respectively.  $\phi_2 v$ 's of sodium starch glycolates have been previously reported as the numbers between 0.005 and 0.045 (Mantovani, et al., 2000) whereas  $\phi_2 v$  of SSG which is chemically identical is 0.031 (Table 2). In addition, the DSC melting traces yield the endotherms closed to 0 °C compared to the exotherms of freezing traces (Figures 2 and 4). Thus the endothermic melting transition of a fully hydrated polymer is used in order to have an appropriate  $\phi_2$ .

Each of  $\phi_2$ - $T$  data sets derived from DSC curves was non-linearly fitted into equation 4 with the restricted conditions that  $(R/\Delta H_m) = 1.383 \times 10^{-3} \text{ K}^{-1}$  and  $T_0 = 273.15 \text{ K}$  (Borchard, et al., 2005). The estimates as well as their standard errors (SE) of parameters including  $\chi_1$ , network factor ( $v_1 \frac{v_e}{V_0}$ ) and  $f$  are tabulated in Table 3. It is noted that ionic and/or cross-linking network contribution factor was set as null for uncharged and/or linear polymers, respectively. It was found that the model is successfully applied to  $\phi_2$ - $T$  data sets with high correlations ( $r^2$ : 0.934-0.999, Table 3). It is thus demonstrated that  $\chi_1$ , charges, and polymer network affect the

crystallization/melting of water that the polymer contains. As seen in Table 3,  $f$ 's of charged polymers are statistically significant from null at  $\alpha$ -level of 0.05, so are network factors of cross-linked ones except CCS.  $f$  reflects the degree of ionization whereas network factor illustrates swelling of the cross-linked polymers (Borchard, et al., 2005; Mantovani, et al., 2000). It is observed that at 0.05- $\alpha$ -level, network factor in the case of CCS is not significantly different from null. It might evidently be because the swelling of the polymeric network is not sufficient to significantly effect on the water crystallization / melting for it was previously reported that the swelling capacity of CCS present in water was far lower than that of SSG (Visavarungroj & Remon, 1990). In addition, Okoroafor, et al. (1998) mentioned that the effect of network factor was quite small since its value usually is of the order of two decimal digits. That is consistent with the current study as it is observed that the estimates of network factor are in the same order of magnitude (Table 3).

#### 4.5 Flory's interaction parameter ( $\chi_1$ )

To characterize the thermodynamic interaction between water and polymer, Flory (1971) introduced a dimensionless quantity:  $\chi_1$ . It represents merely the difference in energy divided by thermal agitation energy ( $kT$ : where  $k$  is Boltzmann's constant) of a solvent molecule immersed in the pure polymer compared with one surrounded by molecules of its own kind. A number of authors reported the magnitudes of  $\chi_1$  of aqueous polymeric solutions including starches and its derivatives (Baks, et al., 2007; Cruz-Orea, et al., 2002; Mantovani, et al., 2000; Farhat & Blanshard, 1997), and sodium alginate (Borchard, et al., 2005) as the numbers ranging between 0.43 and 0.67. As seen in Table 3, the estimates of  $\chi_1$ -parameters of the same types of polymers vary between 0.520 and 0.761 which are comparable. Myagkova, et al. (1997) mentioned that the  $\chi_1$  should be approximately 0.5 for

maximum dissolving capacity of liquid water, i.e., the good-solvent conditions, for cellulose esters whereas the magnitudes of  $\chi_1$  for the same type of polymers under study are 0.679-0.847 which also approaches those conditions. In fact, the magnitude  $\chi_1$  is somewhat empirical and not a constant. It is dependent on volume fraction as well as temperature (Myagkova, et al.; 1997 and Flory, 1971). Thus experimental conditions should affect its magnitude especially during the initial setting causing  $\chi_1$  values to deviate from laboratories to laboratories.

Figure 5 illustrates the plot of  $\chi_1$  versus the reciprocal absolute temperature of the onset of DSC melting transition of water of fraction (ii) in fully hydrated samples. It is observed that the smaller the value of  $\chi_1$ , the larger solvent water melting was depressed, i.e., stronger affinity for water. Flory (1971) rectified the energy quantity of  $\chi_1$  that should be regarded as the free energy change rather than as the heat of mixing only.  $\chi_1$  then contains an entropy contribution in addition to enthalpy one. Thus, in a simple case (Borchard, et al., 2005):

$$(4) \quad \chi_1 = \alpha_1 + \frac{\beta_1}{T}$$

where,  $\alpha_1$  and  $\beta_1$  are entropy and enthalpy parameters, respectively. Assuming the same type of interaction,  $\chi_1$  derived from polymeric hydrogels in this study could exhibit the relationship with  $1/T$  as showed by Equation 4. As seen in Figure 5, the trend line as well as 95% confidence interval (dotted lines in Figure 5) represents the data fitting of Equation 4. Unfortunately, the power of regression and the correlation coefficient are as low as 25.01% and 0.631, respectively. It might be because the variety in nature of individual polymers and experimental conditions could complicate the systems resulting the fitted parameters are so empirical that they are meaningless to address.

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Parameter	DSC run	Reference (1)	Deviation (%)
Onset; Peak (K)	274.15; 275.4	273.15	+0.37; +0.82
Heat of melting (Kcal/mol)	1.453	1.436	+1.18

**Table 1.** Water melting information taken from endotherm (heating trace) compared with the reference (Dean, 1985).

Polymeric material	<sup>1</sup> Overall water content (%)	<sup>2</sup> Freezable water content (%) (mean, s.d.)	<sup>3</sup> Water of fraction (i) (%)	<sup>4</sup> $\phi_{2V}$
PS	80.01	53.38, 1.09	26.63	0.112
SA	73.36	47.43, 1.07	25.93	<sup>a</sup> 0.167
SSG	73.28	52.30, 1.11	20.98	0.031
HPMC	51.30	38.09, 0.87	13.21	<sup>a</sup> 0.384
SCMC	69.13	46.14, 0.78	22.99	0.145
CCS	79.97	70.30, 1.11	9.67	0.054

1. Overall water content was determined by moisture balance.

2. Freezable water content was determined by DSC traces calculation (in 3 replicates) based on the heat of melting in Table 1.

3. Non-freezable water content was calculated as Overall water content minus Freezable water content.

4. The fully hydrated polymer volume fraction based on equation (Mantovani et al., 2000):

$\phi_{2V} = [\overline{d_{al}}/\overline{d_w}]^3$  where  $\overline{d_{al}}$  and  $\overline{d_w}$  are geometric mean volume diameters of a powdered polymer in alcohol and in water, respectively.

a. The numbers are taken from reference (Mutalik et al., 2006) since the equipment could not determine.

**Table 2.** Water contents and the volume fractions of fully hydrated hydrophilic polymers under study.



Polymeric material	$\chi_1$ Estimate, SE	$V_1 \frac{v_e}{V_0}$ Estimate, SE	$f$ Estimate, SE	$r^2$
PS	0.761, 0.041	0.067, 0.017	**_	0.939
SA	0.738, 0.033	*_	0.513, 0.022	0.986
SSG	0.520, 0.051	0.084, 0.010	0.288, 0.093	0.994
HPMC	0.847, 0.032	*_	**_	0.934
SCMC	0.776, 0.021	*_	0.368, 0.070	0.947
CCS	0.679, 0.025	<sup>a</sup> 0.028, 0.048	0.241, 0.002	0.999

\* Since the polymers are linear, network contribution is absent.

\*\* Since the polymers are uncharged, the reduced model with  $f=0$  is used.

<sup>a</sup> The contribution is statistically non-significant at 0.05  $\alpha$ -level.

**Table 3.** The estimates of the parameters according to the restricted ( $R/\Delta H_m = 1.383 \times 10^{-3} \text{ K}^{-1}$  and  $T_0 = 273.15 \text{ K}$ ) non-linear regression of equation 3.

### Figure legends

**Figure 1.** DSC thermograms of water illustrating water crystallization (cooling trace: I) and melting (heating trace: II). DSC was done according to the conditions listed in Table 1.

**Figure 2.** DSC thermograms (cooling [A] and heating [B] curves) of SA previously equilibrated in 100%RH at 30 degrees C for 7 days showing 2 phases of water on a polymer surface. (I) is freezable bound water and (II) is bulk water.

**Figure 3.** DSC freezing traces of water in the samples of CCS equilibrated with (a) 96% RH, (b) 100%RH, and (c) liquid water. It is noted that hydrogels of other polymer under study also exhibit similar behavior.

**Figure 4.** DSC endothermic melting of ice in SSG equilibrated with (a) 84% RH, (b) 96% RH, (c) 100% RH, and (d) excess liquid water (fully hydrated). It is noted that hydrogels of other polymer under study also exhibit similar behavior.

**Figure 5.** The plot of  $\chi_1$ -parameter against the reciprocal of onset temperature (in absolute scale) of melting transition of freezable bound water in water-polymer systems under study.

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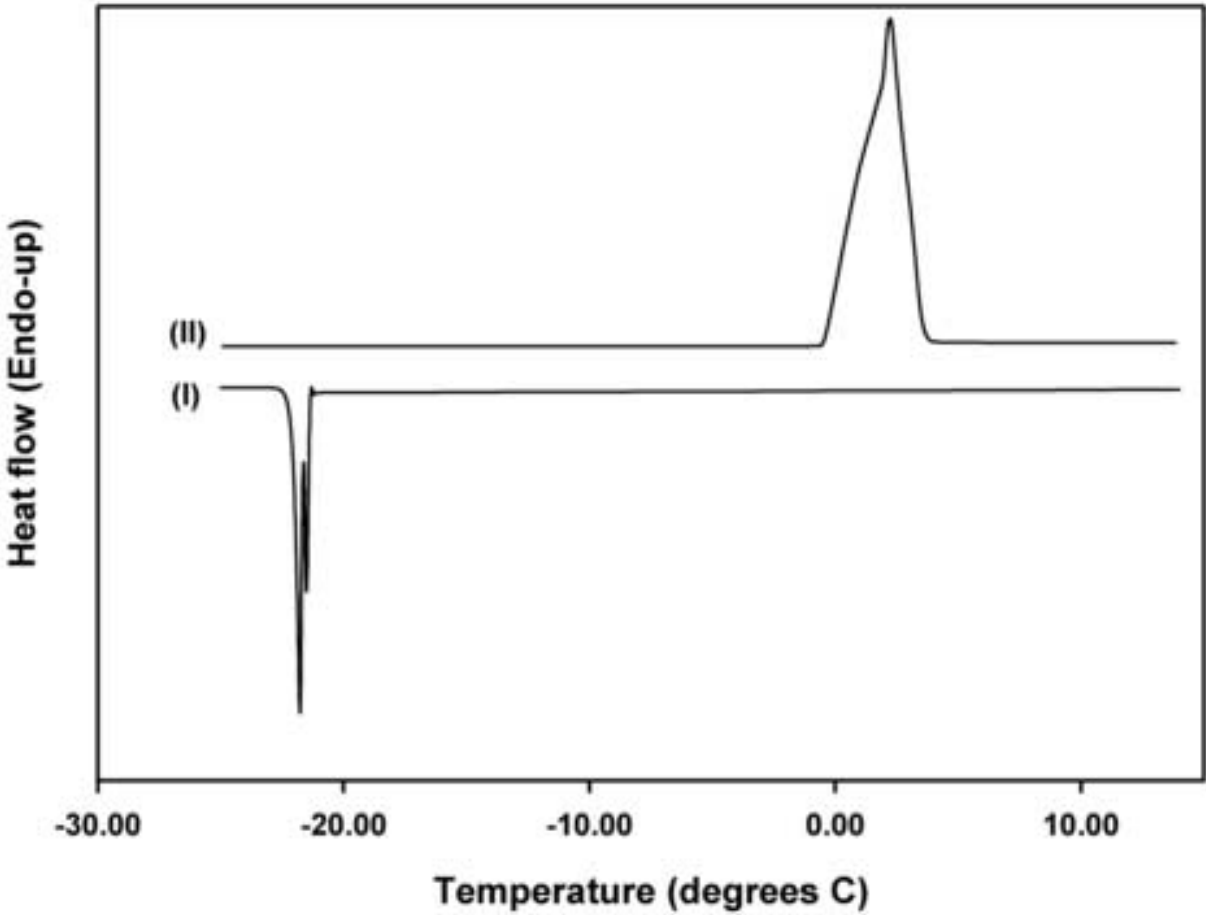


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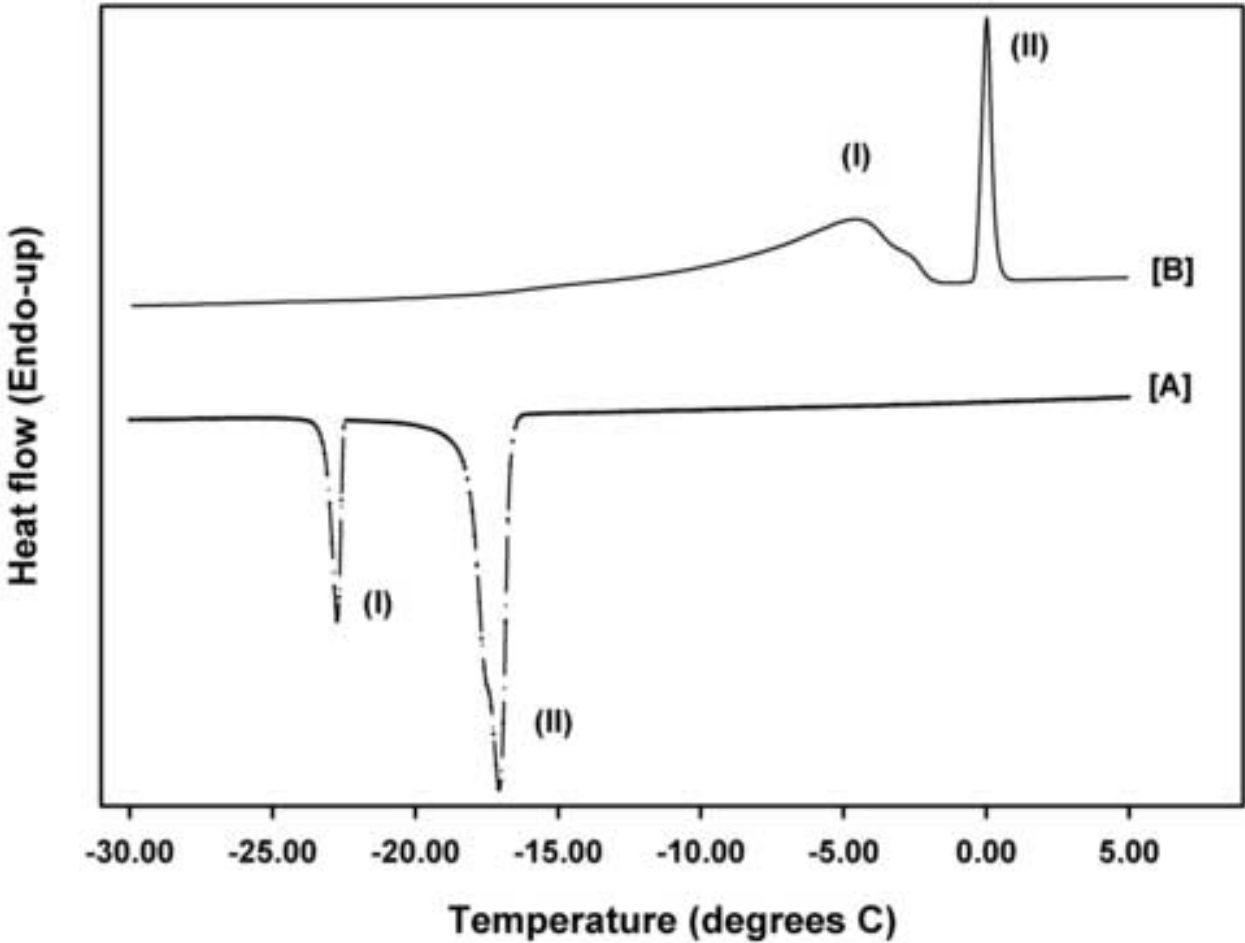


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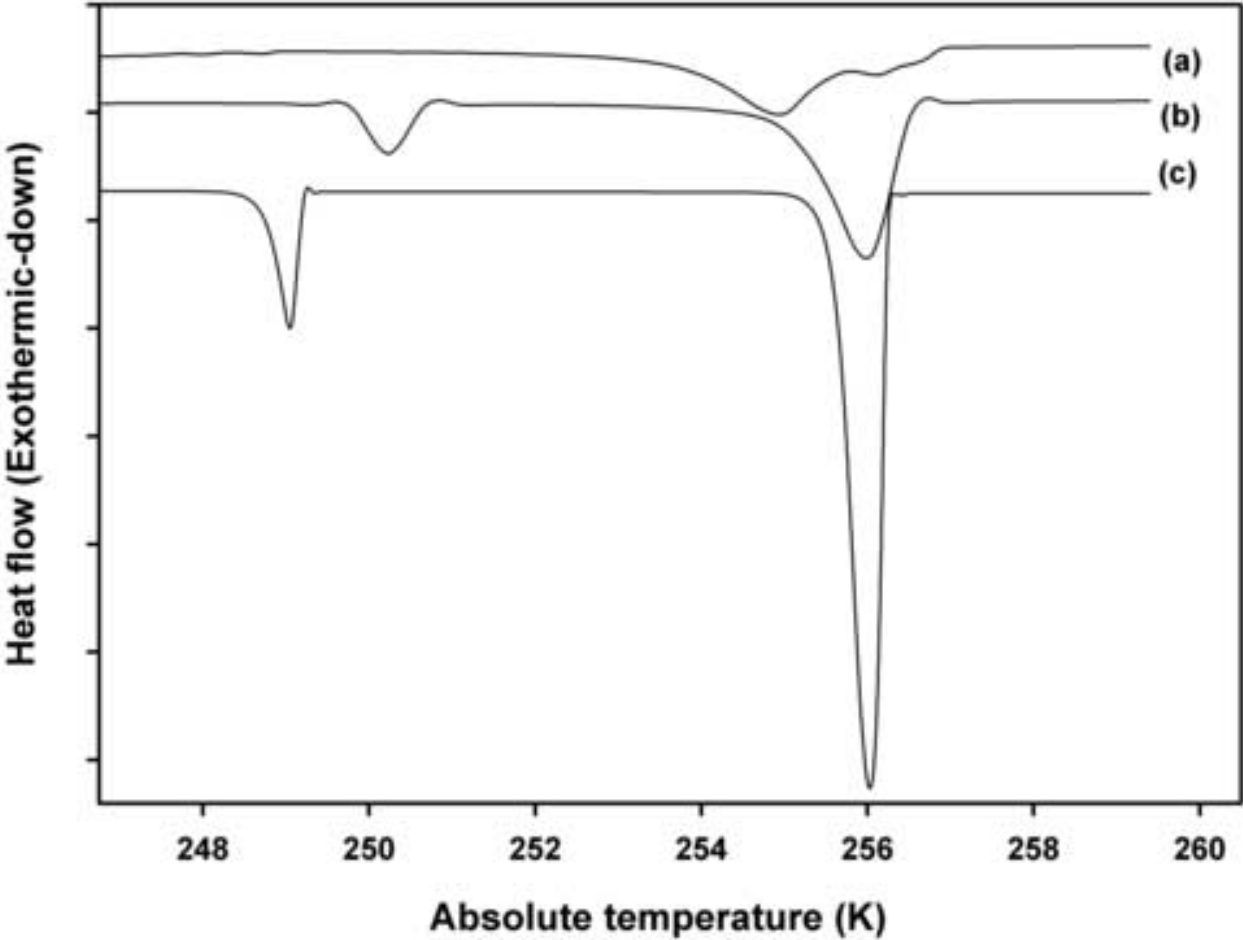


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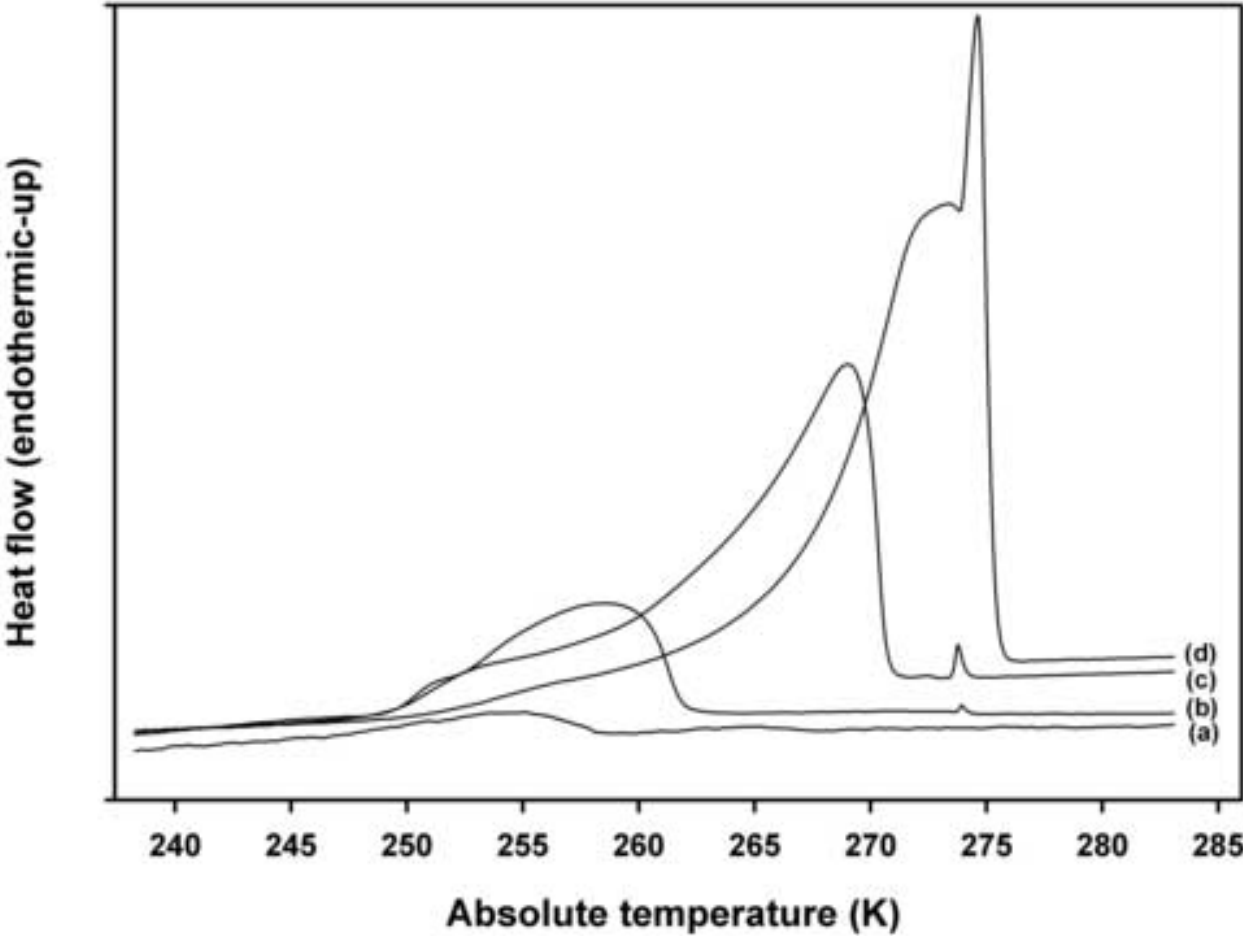
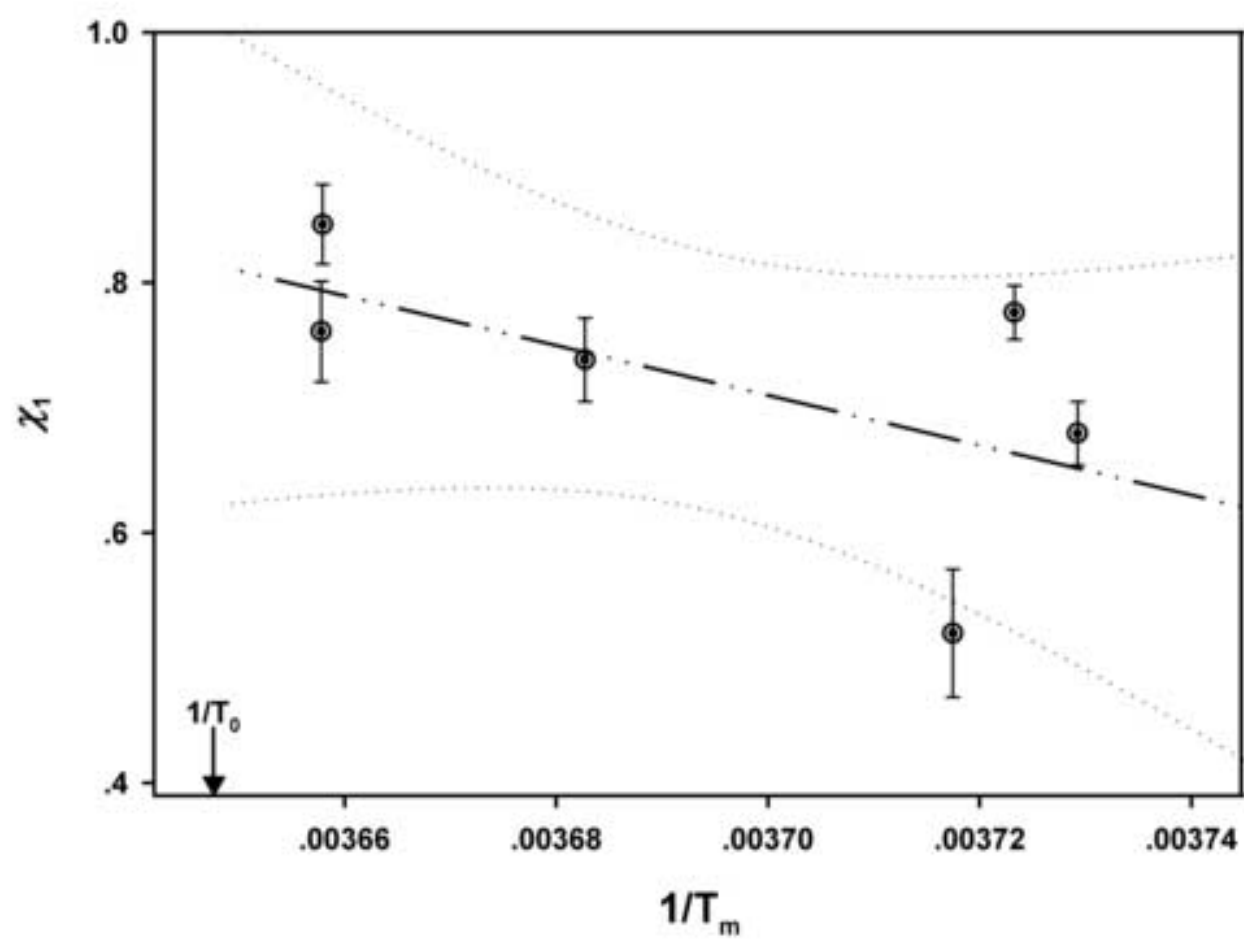


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### **Output** ที่ได้จากโครงการ

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Corresponding Author: Associate Professor Damrongsak - Faroongsarng, Ph.D.

Corresponding Author's Institution: Prince of Songkla University

First Author: Damrongsak - Faroongsarng, Ph.D.

Order of Authors: Damrongsak - Faroongsarng, Ph.D.; Patchara - Sukonrat

Abstract: In a polymer-water matrix, freezable water is depressed due to either porosity confinement or interaction. The aim of the study was to examine water crystallization/melting depression by sub-ambient differential scanning calorimetry. The selected starch and cellulose based excipients including pre-gelatinized starch (PS), sodium alginate, sodium starch glycolate, hydroxypropylmethyl cellulose (HPMC), sodium carboxymethyl cellulose, and croscarmellose sodium were employed. The pre-treated with ambient humidity (85-100% relative humidity, at  $30.0 \pm 0.2^\circ\text{C}$  for 10 days) and with excess water (hydrogels) samples were subjected to a  $25 - -150^\circ\text{C}$ -cooling-heating cycle at  $5.00^\circ\text{C}/\text{min}$  rate. The volume fractions of hydrogels were measured by light scattering technique. It was observed that all polymers but PS and HPMC with ambient humidity presented freezable water in two distinct fractions namely bound water where crystallizing/melting temperature was depressed and bulk water. The water transition in samples with various contents exhibited the pattern as a polymer solution, thus rather than confinement, the depression was due to interaction. The volume fraction-melting temperature data derived from endotherms of hydrogels were successfully fitted to Flory's model ( $r^2$ : 0.934-0.999). The Flory's interaction parameters ( $\chi_1$ ) were found to vary between 0.520 and 0.847. In addition, the smaller the value of  $\chi_1$ , the larger melting was depressed, i.e., stronger affinity for water.

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Faculty of Pharmaceutical Sciences  
Department of Pharmaceutical Technology  
Prince of Songkla University, Hat Yai, Thailand 90112.

June 13, 2007

Dear Editor:

I would like to submit the manuscript entitled: "Thermal behavior of water in polymeric hydrogels of the selected pharmaceutical excipients" to *Int. J. Pharm.* The manuscript describes the crystallization/melting depression of water present in hydrogels of various polymeric pharmaceutical excipients used in drug delivery dosage forms that could be detected by differential scanning calorimetry. We derived the information from DSC tracings and the obtained information was successfully modeled using Flory's polymer solution theory.

Although the DSC technique is not new, the manuscript could provide the novel insight into the thermodynamic principles of polymer-water interaction.

Sincerely,

Damrongsak Faroongsarng, Ph.D.

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**Title:** Thermal behavior of water in polymeric hydrogels of the selected pharmaceutical excipients

**Authors and affiliations:**

\*Damrongsak Faroongsarng  
Drug Delivery Systems Research Center, Department of Pharmaceutical Technology,  
Faculty of Pharmaceutical Sciences, Prince of Songkla University, Hat Yai 90112,  
Thailand.

Patchara Sukonrat  
Scientific Equipment Center, Prince of Songkla University, Hat Yai, 90112, Thailand.

\*Corresponding author  
e-mail: [damrongsak.f@psu.ac.th](mailto:damrongsak.f@psu.ac.th)  
Phone: +66-74-288-841  
FAX: +66-74-428-148

## **Thermal behavior of water in polymeric hydrogels of the selected pharmaceutical excipients**

### **Abstract**

In a polymer-water matrix, freezable water is depressed due to either porosity confinement or interaction. The aim of the study was to examine water crystallization/melting depression by sub-ambient differential scanning calorimetry. The selected starch and cellulose based excipients including pre-gelatinized starch (PS), sodium alginate, sodium starch glycolate, hydroxypropylmethyl cellulose (HPMC), sodium carboxymethyl cellulose, and croscarmellose sodium were employed. The pre-treated with ambient humidity (85-100% relative humidity, at  $30.0 \pm 0.2^\circ\text{C}$  for 10 days) and with excess water (hydrogels) samples were subjected to a  $25 - -150^\circ\text{C}$ -cooling-heating cycle at  $5.00^\circ\text{C}/\text{min}$  rate. The volume fractions of hydrogels were measured by light scattering technique. It was observed that all polymers but PS and HPMC with ambient humidity presented freezable water in two distinct fractions namely bound water where crystallizing/melting temperature was depressed and bulk water. The water transition in samples with various contents exhibited the pattern as a polymer solution, thus rather than confinement, the depression was due to interaction. The volume fraction-melting temperature data derived from endotherms of hydrogels were successfully fitted to Flory's model ( $r^2$ : 0.934-0.999). The Flory's interaction parameters ( $\chi_1$ ) were found to vary between 0.520 and 0.847. In addition, the smaller the value of  $\chi_1$ , the larger melting was depressed, i.e., stronger affinity for water.

*Keywords:* freezable water, water-polymer interaction, Flory's interaction parameter ( $\chi_1$ ), melting point depression, hydrogel.

## 1. Introduction

Starch and cellulose based materials derived from naturally occur biopolymers are the major pharmaceutical excipients utilized in drug delivery dosage forms. These polymers always interact with water due to their hydrophilicity exhibiting some properties that may critically affect the dosage form performance. For example: In controlled release devices, water diffusion through a polymeric hydrogel layer has been considered as one of the major factors determining drug release rate (Rajabi-Siahboomi, et al., 1996). With liquid water in excess, these hydrophilic polymers could form hydrogels i.e., the three-dimensional arrangement possessing the ability to retain a significant fraction of water without complete dissolving. A hydrogel might form relatively stable space lattice or network pores fulfilled with a considerable amount of water. The interfacial tension related to surface of curvature of water within pores could develop and affect the phase transition of the water. Thus this phase transition of water confinement could somehow characterize the pores where it occupies. A number of authors, for examples: Yamamoto et al. (2005), Faroongsarn & Peck (2003), Hay & Laity (2000), and Ishikiriyama & Todoki (1995) examined the pore sizes and distributions of various porous materials assuming that water is mostly held within pores, with melting temperature being depressed by Gibbs-Thomson effect. However, the depression of melting temperature is not only attributed by water confinement in porosity but the water-polymer interaction. Rault et al (1994) reported that the melting depression and the concentration of unfrozen water varied with the water concentration with similar orders of magnitude for polymer-water systems and simple binary mixtures, presenting the same type of interaction, from which confinement effects are absent. They concluded that the melting depression is due not to water confinement in polymer network porosity but rather to water-polymer interactions. The evidence was later confirmed by the work of Okoroafor et al (1998).

In general, interactions between macromolecules fall into four categories: ionic, hydrophobic, van der Waals and hydrogen bonding (Ilmain, et al., 1991). But for a polymer-water mixture, the interaction is always in the range of hydrogen bonding. It has been proposed (Ping, et al., 2001; Zografi & Kontny, 1986; and Higuchi & Iijima, 1985) that water in hydrophilic polymer matrices presents in three distinct fractions: (i) non-freezable bound water, (ii) freezable bound water, and (iii) free or bulk water. Upon cooling, water begins to crystallize only when its content is above a characteristic threshold. This fraction of water has been called freezable bound water (fraction (ii)) because it exhibits a melting point lower than zero °C which is distinguished from bulk water and it should correspond to the depression phenomenon described above. In the lower-than-threshold level, i.e., the water of fraction (i), the molecules of water interact with polar functional groups such as carboxyl groups on polymer chains. The interaction would be well-oriented hydrogen bonding which is locally favorable configuration that being strong enough to prevent water to form ice crystals (Ping, et al., 2001). The differential scanning calorimetric (DSC) study can reveal the freezable water fractions, for example: Nakamura et al. (1981) reported two DSC peaks of crystallization of absorbed water on celluloses including a broad peak observed at ~230-250 K and a sharp one at ~255 K. Should the melting depression of water of fraction (ii) be due to polymer-water interaction, the corresponding DSC peak then could describe the thermodynamics of a polymer-water system. Many techniques are available for the experimental determination of the interaction parameter between solvent molecules and the polymeric chain segment. However, the methods were usually based on volumetric determinations (Mantovani, et al., 2000). The melting/freezing depression determined by DSC could also exhibit the great potential to characterize that interaction. The aim of the study is to examine



the thermal behavior of water melting depression due to its interaction with the selected starch and cellulose based polymers commonly used in drug delivery formulations by mean of DSC technique.

## 2. The thermodynamic relations for a polymer solution

A general thermodynamic theory of polymer solution based on mixing according to liquid lattice theory has been presented by Flory (1971). For polymeric hydrogels employed in the present study, the chemical potential of water ( $\mu_1$ ) in a water-polymer system includes not only Flory's mixing with swollen gel but the Donnan equilibrium for polyelectrolytes that yields the following relationship (Flory, 1971; Okoroafor, et al., 1998; Mantovani, et al., 2000; Ozmen & Okay, 2005; and Borchard, et al., 2005):

$$(1) \quad \mu_1^{gel} - \mu_1^0 = RT[\ln(1 - \phi_2) + \phi_2 + \chi_1 \phi_2^2 + V_1(\frac{v_e}{V_0})(\phi_2^{1/3} - \frac{\phi_2}{2}) - f \cdot \phi_2]$$

Where,  $\phi_2$ ,  $\chi_1$ ,  $v_e$ , and  $f$  are volume fraction of polymer in gel, the Flory's polymer-water interaction parameter, the effective crosslink density of the network, and the fraction of charged units in the hydrogel network, respectively.  $V_1$  and  $V_0$  are molar volume of water and the volume of relaxed hydrogel network.  $R$  and  $T$  are gas constant and absolute temperature.  $\mu_1^0$  is the chemical potential of pure liquid water. And,  $\mu_1^{gel}$  is the chemical potential of water in hydrogel. The first three terms in the right hand side of equation 1 represent the chemical potential of general polymer-water mixture. The fourth term is the chemical potential due to reaction of the network crosslink structure (Flory, 1971), whereas the last term is that from Donnan equilibrium theory (Mantovani et al. 2000; and Ozmen & Okay, 2005).

It is further assumed that frozen water is in equilibrium with the unfrozen water in gel phase during the DSC operation, i.e., the chemical potential of freezing

ice ( $\mu_1^{ice}$ ) and of water in hydrogel ( $\mu_1^{gel}$ ) must be equal. And when a mixture freezes, one of the colligative properties known as freezing point depression holds. The change of chemical potential can be written as (Ozmen & Okay, 2005):

$$(2) \quad \mu_1^{ice} - \mu_1^0 = \Delta H_m \left( \frac{T}{T_0} - 1 \right)$$

Where  $\Delta H_m$ , and  $T_0$  are molar enthalpy of crystallization (or melting), and melting temperature of pure water, respectively. Since the left hand side of equation 1 and 2 are equal, the arrangement of these two equations yields:

$$(3) \quad \frac{1}{T} = \frac{1}{T_0} - \frac{R}{\Delta H_m} \cdot [\ln(1 - \varphi_2) + \varphi_2 + \chi_1 \varphi_2^2 + V_1 \left( \frac{V_e}{V_0} \right) (\varphi_2^{1/3} - \frac{\varphi_2}{2}) - f \cdot \varphi_2]$$

This equation should be applicable to the water of fraction (ii) where the ice-liquid water transition temperature was depressed. And, assuming the involved parameters are constant over the transition temperature, the parameters such as  $\chi_1$  could be obtained by non-linear regression of  $\frac{1}{T}$  as a function of  $\varphi_2$  according to the model described by equation 3.

### 3. Materials and Method

#### 3.1 Materials

The variety in nature of starch and cellulose based polymers including pre-gelatinized potato starch (PS: Starch<sup>®</sup> 1500, Colorcon, Inc., PA, USA), sodium alginate (SA: Wendt-Chemie, Hamburg, Germany), sodium starch glycolate (SSG: Explotab<sup>®</sup>, JRS Pharma, Rosenberg, Germany), hydroxypropyl methyl cellulose (HPMC: Colorcon, Inc., PA, USA), Sodium carboxymethyl cellulose (SCMC: Wendt-Chemie, Hamburg, Germany) and croscarmellose sodium (CCS: Ac-di-sol<sup>®</sup>, FMC Corp. PA, USA) were employed. SA, PS, and SSG were charged-linear, branch

and linear, and charged-crosslinked polysaccharides, respectively. HPMC, SCMC, and CCS were linear, charged-linear, and charged-crosslinked celluloses, respectively.

### 3.2 *Sub-ambient differential scanning calorimetric study*

The Perkin-Elmer differential scanning calorimeter (DSC7 with TAC7/DX Thermal analysis controller, Perkin-Elmer Corp., Norwalk, CT, USA) equipped with liquid nitrogen bath set as a cooling accessory was employed. Calibrations with Indium and cyclohexane were carried out for every time which the DSC operation started to ensure the accuracy/precision of the obtained heat of transitions and the corresponding temperatures. An accurately weighed (5-15 mg) sample was placed in tightly sealed aluminum pan (Perkin-Elmer Corp., Norwalk, CT, USA). The samples were subjected to run against an empty pan as a reference. With loading temperature of 25 °C, the analysis program includes 1) cooling from 25 °C to -150 °C at 5.00 °C/min rate, 2) isothermal run at -150 °C for 1 min, and 3) heating from -150 °C to 25 °C at the same rate as cooling step. The distilled water was run to validate the temperature and heat of water crystallization/melting. All of DSC thermograms (cooling or heating traces) were analyzed using Pyris<sup>®</sup> software (Perkin-Elmer Perkin-Elmer Corp., Norwalk, CT, USA).

The samples were pre-treated with ambient humidity prior to DSC analyses. The ~5 g-samples were equilibrated with 85, 96, and 100% relative humidity (RH) at 30.0±0.2 °C for 10 days. The samples were also fully hydrated by liquid water in excess at the same temperature as those pre-treated with ambient humidity as follows: A 3- to 8-gram sample (equivalent to approximately 10-ml bulk volume) was thoroughly mixed with liquid water to 100 ml in volume. The mixtures were allowed to be still for 1 day. Hydrogels or sediments depending to the nature of water-polymer mixtures were subjected to sub-ambient differential scanning calorimetric study

described above. The total water ( $W_T$ ) contents of hydrogel/sediment samples were determined using a moisture balance (Mettler® LP16 & PM300, Mettler-Toledo, Inc., Hightstown, NJ, USA) with heating temperature of 100 °C.

### 3.3 The determination of non-freezable water

The water of fraction (i) was calculated by subtracting the total water content ( $W_T$ ) by the water content calculated from the amount of heat corresponded to DSC melting traces in sub-ambient temperatures assuming that the area of melting peak of pure water corresponds to the melting enthalpy. So, the heat was converted to the amount of water since it was directly proportional to enthalpy of melting obtained from DSC tracing of distilled water.

### 3.4 The determination of polymer volume fraction in liquid water

The fully hydrated polymer volume fraction ( $\phi_{2V}$ ) was obtained from particle size determination in non-swelling and swelling states, as analogous to what was done previously (Mantovani et al., 2000). The size and distribution of each of the polymeric powders were measured by dynamic laser light scattering technique (Mastersizer®/E, Malvern Instruments Ltd., Worcestershire, UK). Alcohol and water were used as non-swelling and swelling media, respectively.  $\phi_{2V}$  was obtained by comparing mean volume diameters according to the equation of  $\phi_{2V} = [\overline{d_{al}}/\overline{d_w}]^3$ , where  $\overline{d_{al}}$  and  $\overline{d_w}$  are geometric mean volume diameters of a powdered polymer in alcohol and in water, respectively.

To quantify the polymer volume fraction during ice-liquid phase transition of water denoted by  $\phi_2$ , it was assumed that only pure water freezes when cooled to freezing point.  $\phi_2$  is thus directly proportional to the cumulative partial area under the DSC peak at corresponding  $T$ , i.e.,  $\phi_2 = \phi_2^{(i)} - \Lambda \frac{AT}{P}$ . Where,  $A$ ,  $T$ ,  $P$ ,  $\phi_2^{(i)}$ , and  $\Lambda$  are

the area under the peak at temperature  $T$ , the total area under the peak, the polymer volume fraction with water of fraction (i), and the linear coefficient that makes  $\phi_2$  equals  $\phi_{2V}$  determined by light scattering technique, in which  $A_T$  equals  $P$ , respectively.  $\phi_2^{(i)}$  was approximated from mole fraction of water of fraction (i) ( $x_1^{(i)}$ ) calculated based on the water content of non-freezable water previously described, i.e.,  $\phi_2^{(i)} \approx (1 - x_1^{(i)})$ . The  $\phi_2$  and its corresponding  $T$  were non-linearly fitted into Flory's model using the commercial software (SigmaPlot® 2000, SPSS, Inc.).

#### 4. Results and Discussion

##### 4.1 *In situ* water crystallization: the validation of DSC measurement

The cooling and heating traces revealing water crystallization and melting, respectively, are in Figure 1. There was an exothermic peak of water crystallization (I in Figure 1) occurred at a temperature far below zero °C. Endothermic melting peak (II in Figure 1), on the other hand, started at a normal melting temperature. This inconsistency between freezing and melting curves is commonly observed in fairly slow rate of scanning (1-10 °C/min). It is because the crystallization difficulty causes an exotherm to appear at a temperature lower than normal. It seems that the melting trace could approach an equilibrium ice-water transition better than cooling counterpart as the tracing was close to 0 °C. Table 1 shows the detailed information of water melting (II in Figure 1) compared with the reference (Dean, 1985).

As seen in Table 1, both onset and heat of melting for pure water agree with the values taken from the reference. Very low deviations, i.e., 0.37% and 1.18% deviate from reference values for onset and heat of melting, respectively, are observed. It has been stated that in typical DSC measurement, the mean error at heating/cooling rate of 1-10 °C/min should not exceed 2.5% (Borchard, et al., 2005).

Thus, the method and its conditions could be used to investigate water crystallization/melting with acceptable precision and accuracy.

#### *4.2 DSC water tracings in the selected hydrophilic polymers and the nature of ice-liquid water transition*

Figure 2 illustrates the tracings of water that could be found in SA, SSG, SCMC, and CCS equilibrated with ambient humidity (85-100% RH). For simplicity, the only tracings of SA-100%RH system are showed. As seen in Figure 2, the freezable water in current study is consistent with previous report (Nakamura et al., 1981). It is then subjected into 2 fractions, i.e., water of fraction (ii) labeled as (I) where freezing/melting happen at a temperature below zero, and that of fraction (iii) labeled as (II) where its transitions are closed to normal melting point. Figure 3 illustrates the DSC freezing traces of CCS with various aqueous level environments including that with liquid water in excess. It is noted that other polymers in this study showed similar patterns. However, the water transition tracings were absent in the cases of PS and HPMC in ambient humidity but fully hydrated samples. PS and HPMC are non-ionic polymers exhibiting less hygroscopic than others. It may be because ionic species and salts could attribute to hydration on polymer they present with and might allow amount of water uptake greater than threshold of non-frozen water to show the DSC tracings of water of fraction (ii) and (iii) in cases of SA, SSG, SCMC, and CCS.

Should the porosity formed by 3-dimentional polymer network govern the freezing/melting point depression, the depressed temperature in various moisture environments of the same polymer which would form similar pore structures might be invariant. Furthermore, if the pores collapse during ice formation, the transition of water of fraction (ii) might be either near or far from that of water of fraction (iii) dependent on the new size of the pores that water occupies after collapsing. As

obviously showed in Figures 3, there are not the cases in the present study. It is observed that the phase transition of water of fraction (ii) always exhibits a pattern as a polymer solution, i.e., the more concentration level of water; the more freezing temperature is depressed. Thus rather than porosity confinement, the freezing temperature may be depressed in accord with polymer-water interaction.

Figure 4 illustrates the endothermic melting traces of SSG with variety of humidity as well as fully hydrated sample. Like freezing exotherms, the melting endotherms of various level of water with polymer samples under study were also in similar patterns. It is observed that the melting of freezable bound water shifts toward the melting of free water. i.e., the two singlet peaks turn to a doublet with increase in water content which is similar to the previous study (Borchard et al., 2005). It may be because water of fraction (ii), during increasing temperature, becoming liquid phase migrates from the vicinity of polymer interaction sites within gel due to hydrogen bonding among water molecules to be in equilibrium again with free water that melt later at a normal melting temperature.

#### 4.3 *Non-freezable bound water*

An attempt at the determination of water of fraction (i) for each of polymer-water systems was made and tabulated in Table 2. The materials under study exhibit the non-freezable water contents of between 9.67% and 26.63% whereas it was previously reported that starches and celluloses exhibited non-freezable bound water contents of 28% (Zhong & Sun, 2005) and 22-25% (Luukkonen, et al., 2001), respectively. McCrystal, et al (1997) estimated the number of moles of non-freezing water per a polymer repeating unit for HPMC gel as approximately 3.8 moles that is corresponding to approximately 10-20% water content dependent on degrees of substitution, while the current study on HPMC is within the range (13.21%, Table 2).

On the other hand, the cross-linked chemically modified starch and cellulose that are more hygroscopic (SSG and CCS) illustrate low level of non-freezable bound water (Table 2). It might be because these materials present more number of ice nuclei, during freezing, that draw more water molecules due to hydrogen bonding to the ice clusters as a process of lowering surface free energy. As a result more portion of freezable water may be detected.

#### 4.4 *The volume fraction of polymeric hydrogels vs. melting depression: non-linear fitting to the Flory's model*

The volume fractions in liquid water ( $\phi_2 v$ 's) of fully hydrated polymers under study are tabulated in Table 2. It is noted that  $\phi_2 v$ 's of SA and HPMC have been taken from the reference (Mutalik, et al., 2006) since the polymers dissolved in water and alcohol, respectively.  $\phi_2 v$ 's of sodium starch glycolates have been previously reported as the numbers between 0.005 and 0.045 (Mantovani, et al., 2000) whereas  $\phi_2 v$  of SSG which is chemically identical is 0.031 (Table 2). In addition, the DSC melting traces yield the endotherms closed to 0 °C compared to the exotherms of freezing traces (Figures 2 and 4). Thus the endothermic melting transition of a fully hydrated polymer is used in order to have an appropriate  $\phi_2$ .

Each of  $\phi_2$ - $T$  data sets derived from DSC curves was non-linearly fitted into equation 4 with the restricted conditions that  $(R/\Delta H_m) = 1.383 \times 10^{-3} \text{ K}^{-1}$  and  $T_0 = 273.15 \text{ K}$  (Borchard, et al., 2005). The estimates as well as their standard errors (SE) of parameters including  $\chi_1$ , network factor ( $v_1 \frac{v_e}{V_0}$ ) and  $f$  are tabulated in Table 3. It is noted that ionic and/or cross-linking network contribution factor was set as null for uncharged and/or linear polymers, respectively. It was found that the model is successfully applied to  $\phi_2$ - $T$  data sets with high correlations ( $r^2$ : 0.934-0.999, Table 3). It is thus demonstrated that  $\chi_1$ , charges, and polymer network affect the



crystallization/melting of water that the polymer contains. As seen in Table 3,  $f$ 's of charged polymers are statistically significant from null at  $\alpha$ -level of 0.05, so are network factors of cross-linked ones except CCS.  $f$  reflects the degree of ionization whereas network factor illustrates swelling of the cross-linked polymers (Borchard, et al., 2005; Mantovani, et al., 2000). It is observed that at 0.05- $\alpha$ -level, network factor in the case of CCS is not significantly different from null. It might evidently be because the swelling of the polymeric network is not sufficient to significantly effect on the water crystallization / melting for it was previously reported that the swelling capacity of CCS present in water was far lower than that of SSG (Visavarungroj & Remon, 1990). In addition, Okoroafor, et al. (1998) mentioned that the effect of network factor was quite small since its value usually is of the order of two decimal digits. That is consistent with the current study as it is observed that the estimates of network factor are in the same order of magnitude (Table 3).

#### 4.5 Flory's interaction parameter ( $\chi_1$ )

To characterize the thermodynamic interaction between water and polymer, Flory (1971) introduced a dimensionless quantity:  $\chi_1$ . It represents merely the difference in energy divided by thermal agitation energy ( $kT$ : where  $k$  is Boltzmann's constant) of a solvent molecule immersed in the pure polymer compared with one surrounded by molecules of its own kind. A number of authors reported the magnitudes of  $\chi_1$  of aqueous polymeric solutions including starches and its derivatives (Baks, et al., 2007; Cruz-Orea, et al., 2002; Mantovani, et al., 2000; Farhat & Blanshard, 1997), and sodium alginate (Borchard, et al., 2005) as the numbers ranging between 0.43 and 0.67. As seen in Table 3, the estimates of  $\chi_1$ -parameters of the same types of polymers vary between 0.520 and 0.761 which are comparable. Myagkova, et al. (1997) mentioned that the  $\chi_1$  should be approximately 0.5 for

maximum dissolving capacity of liquid water, i.e., the good-solvent conditions, for cellulose esters whereas the magnitudes of  $\chi_1$  for the same type of polymers under study are 0.679-0.847 which also approaches those conditions. In fact, the magnitude  $\chi_1$  is somewhat empirical and not a constant. It is dependent on volume fraction as well as temperature (Myagkova, et al.; 1997 and Flory, 1971). Thus experimental conditions should affect its magnitude especially during the initial setting causing  $\chi_1$  values to deviate from laboratories to laboratories.

Figure 5 illustrates the plot of  $\chi_1$  versus the reciprocal absolute temperature of the onset of DSC melting transition of water of fraction (ii) in fully hydrated samples. It is observed that the smaller the value of  $\chi_1$ , the larger solvent water melting was depressed, i.e., stronger affinity for water. Flory (1971) rectified the energy quantity of  $\chi_1$  that should be regarded as the free energy change rather than as the heat of mixing only.  $\chi_1$  then contains an entropy contribution in addition to enthalpy one. Thus, in a simple case (Borchard, et al., 2005):

$$(4) \quad \chi_1 = \alpha_1 + \frac{\beta_1}{T}$$

where,  $\alpha_1$  and  $\beta_1$  are entropy and enthalpy parameters, respectively. Assuming the same type of interaction,  $\chi_1$  derived from polymeric hydrogels in this study could exhibit the relationship with  $1/T$  as showed by Equation 4. As seen in Figure 5, the trend line as well as 95% confidence interval (dotted lines in Figure 5) represents the data fitting of Equation 4. Unfortunately, the power of regression and the correlation coefficient are as low as 25.01% and 0.631, respectively. It might be because the variety in nature of individual polymers and experimental conditions could complicate the systems resulting the fitted parameters are so empirical that they are meaningless to address.

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Parameter	DSC run	Reference (1)	Deviation (%)
Onset; Peak (K)	274.15; 275.4	273.15	+0.37; +0.82
Heat of melting (Kcal/mol)	1.453	1.436	+1.18

**Table 1.** Water melting information taken from endotherm (heating trace) compared with the reference (Dean, 1985).

Polymeric material	<sup>1</sup> Overall water content (%)	<sup>2</sup> Freezable water content (%) (mean, s.d.)	<sup>3</sup> Water of fraction (i) (%)	<sup>4</sup> $\phi_{2V}$
PS	80.01	53.38, 1.09	26.63	0.112
SA	73.36	47.43, 1.07	25.93	<sup>a</sup> 0.167
SSG	73.28	52.30, 1.11	20.98	0.031
HPMC	51.30	38.09, 0.87	13.21	<sup>a</sup> 0.384
SCMC	69.13	46.14, 0.78	22.99	0.145
CCS	79.97	70.30, 1.11	9.67	0.054

1. Overall water content was determined by moisture balance.

2. Freezable water content was determined by DSC traces calculation (in 3 replicates) based on the heat of melting in Table 1.

3. Non-freezable water content was calculated as Overall water content minus Freezable water content.

4. The fully hydrated polymer volume fraction based on equation (Mantovani et al., 2000):

$\phi_{2V} = [\overline{d_{al}}/\overline{d_w}]^3$  where  $\overline{d_{al}}$  and  $\overline{d_w}$  are geometric mean volume diameters of a powdered polymer in alcohol and in water, respectively.

a. The numbers are taken from reference (Mutalik et al., 2006) since the equipment could not determine.

**Table 2.** Water contents and the volume fractions of fully hydrated hydrophilic polymers under study.

Polymeric material	$\chi_1$ Estimate, SE	$V_1 \frac{v_e}{V_0}$ Estimate, SE	$f$ Estimate, SE	$r^2$
PS	0.761, 0.041	0.067, 0.017	**_	0.939
SA	0.738, 0.033	*_	0.513, 0.022	0.986
SSG	0.520, 0.051	0.084, 0.010	0.288, 0.093	0.994
HPMC	0.847, 0.032	*_	**_	0.934
SCMC	0.776, 0.021	*_	0.368, 0.070	0.947
CCS	0.679, 0.025	<sup>a</sup> 0.028, 0.048	0.241, 0.002	0.999

\* Since the polymers are linear, network contribution is absent.

\*\* Since the polymers are uncharged, the reduced model with  $f=0$  is used.

<sup>a</sup> The contribution is statistically non-significant at 0.05  $\alpha$ -level.

**Table 3.** The estimates of the parameters according to the restricted ( $R/\Delta H_m = 1.383 \times 10^{-3} \text{ K}^{-1}$  and  $T_0 = 273.15 \text{ K}$ ) non-linear regression of equation 3.

### Figure legends

**Figure 1.** DSC thermograms of water illustrating water crystallization (cooling trace: I) and melting (heating trace: II). DSC was done according to the conditions listed in Table 1.

**Figure 2.** DSC thermograms (cooling [A] and heating [B] curves) of SA previously equilibrated in 100%RH at 30 degrees C for 7 days showing 2 phases of water on a polymer surface. (I) is freezable bound water and (II) is bulk water.

**Figure 3.** DSC freezing traces of water in the samples of CCS equilibrated with (a) 96% RH, (b) 100%RH, and (c) liquid water. It is noted that hydrogels of other polymer under study also exhibit similar behavior.

**Figure 4.** DSC endothermic melting of ice in SSG equilibrated with (a) 84% RH, (b) 96% RH, (c) 100% RH, and (d) excess liquid water (fully hydrated). It is noted that hydrogels of other polymer under study also exhibit similar behavior.

**Figure 5.** The plot of  $\chi_1$ -parameter against the reciprocal of onset temperature (in absolute scale) of melting transition of freezable bound water in water-polymer systems under study.



Figure 1  
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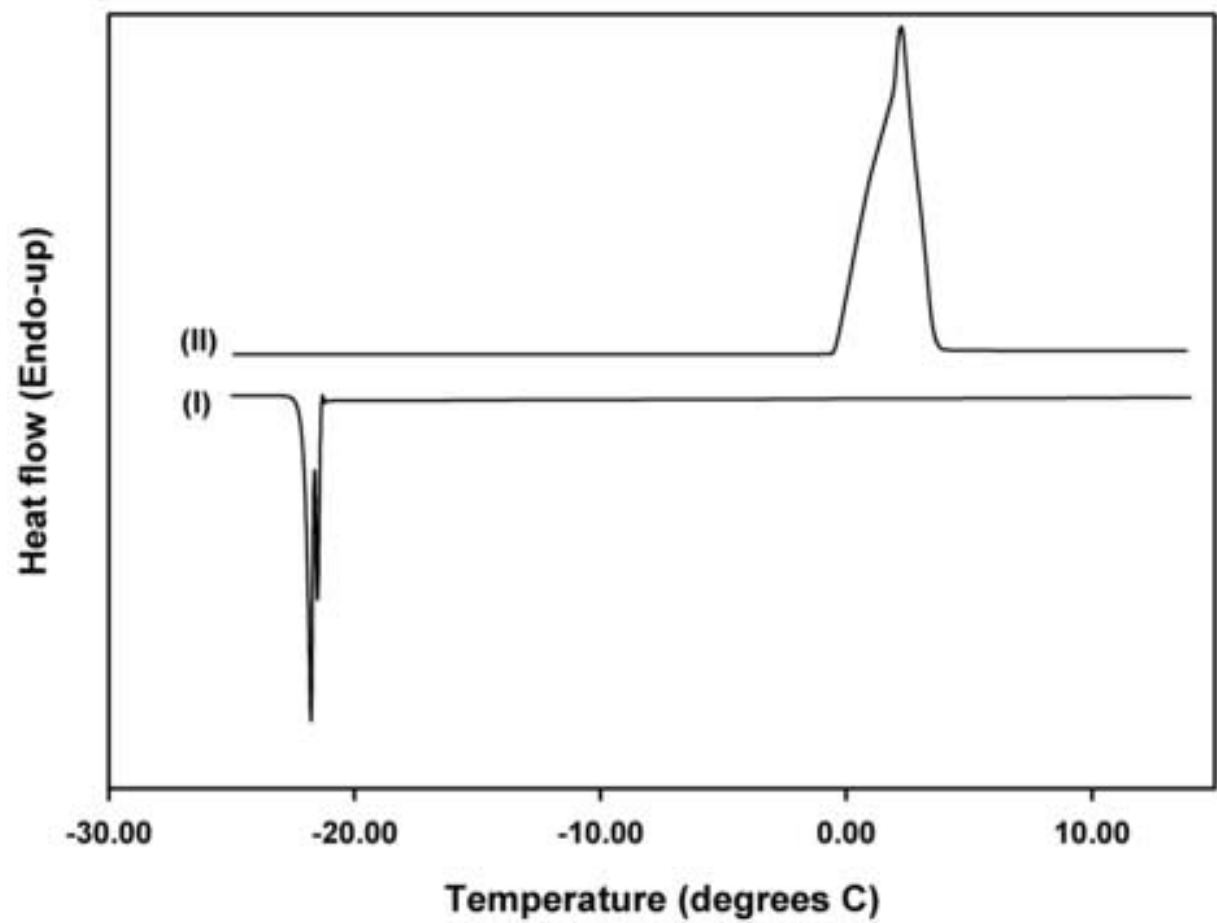


Figure 2  
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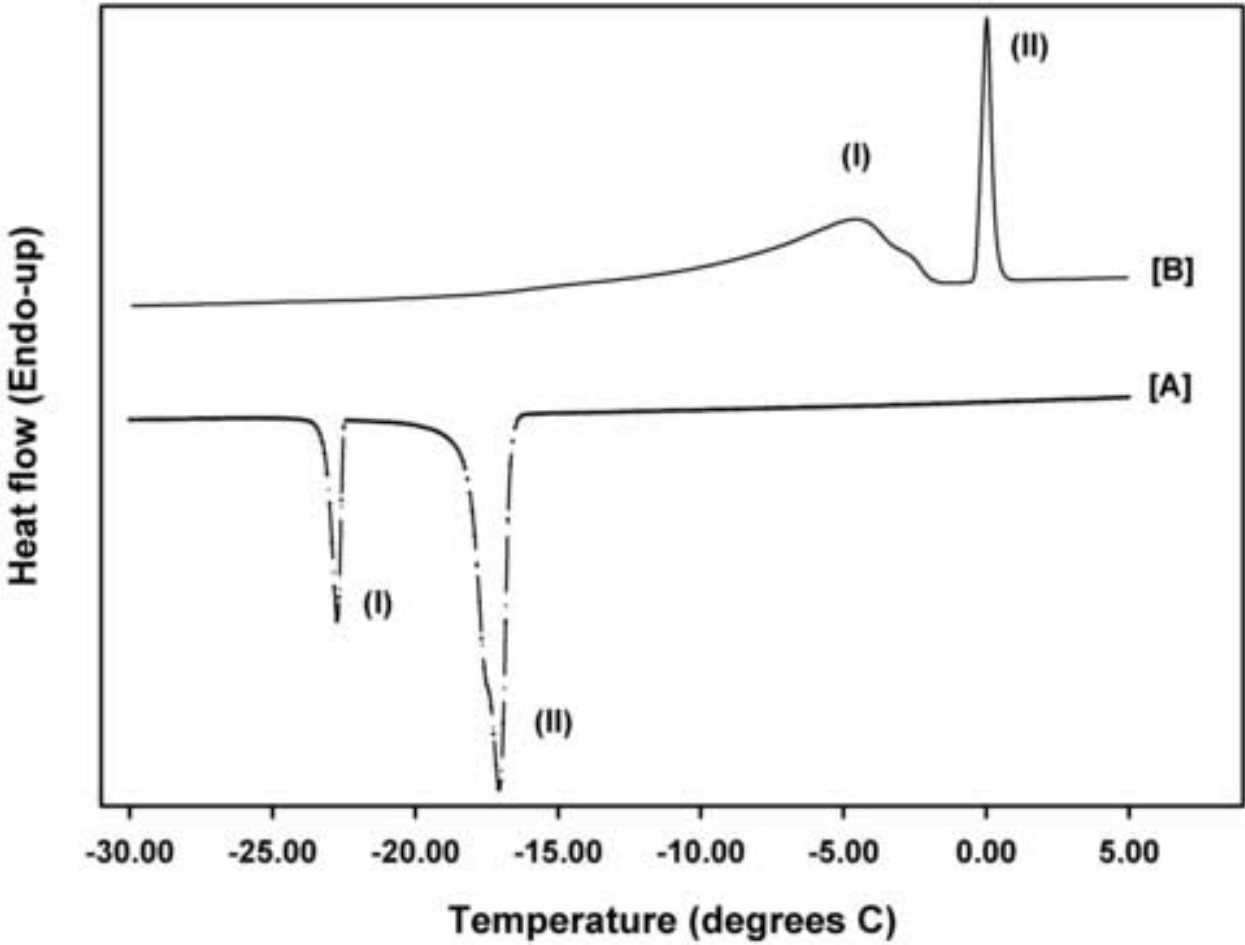


Figure 3  
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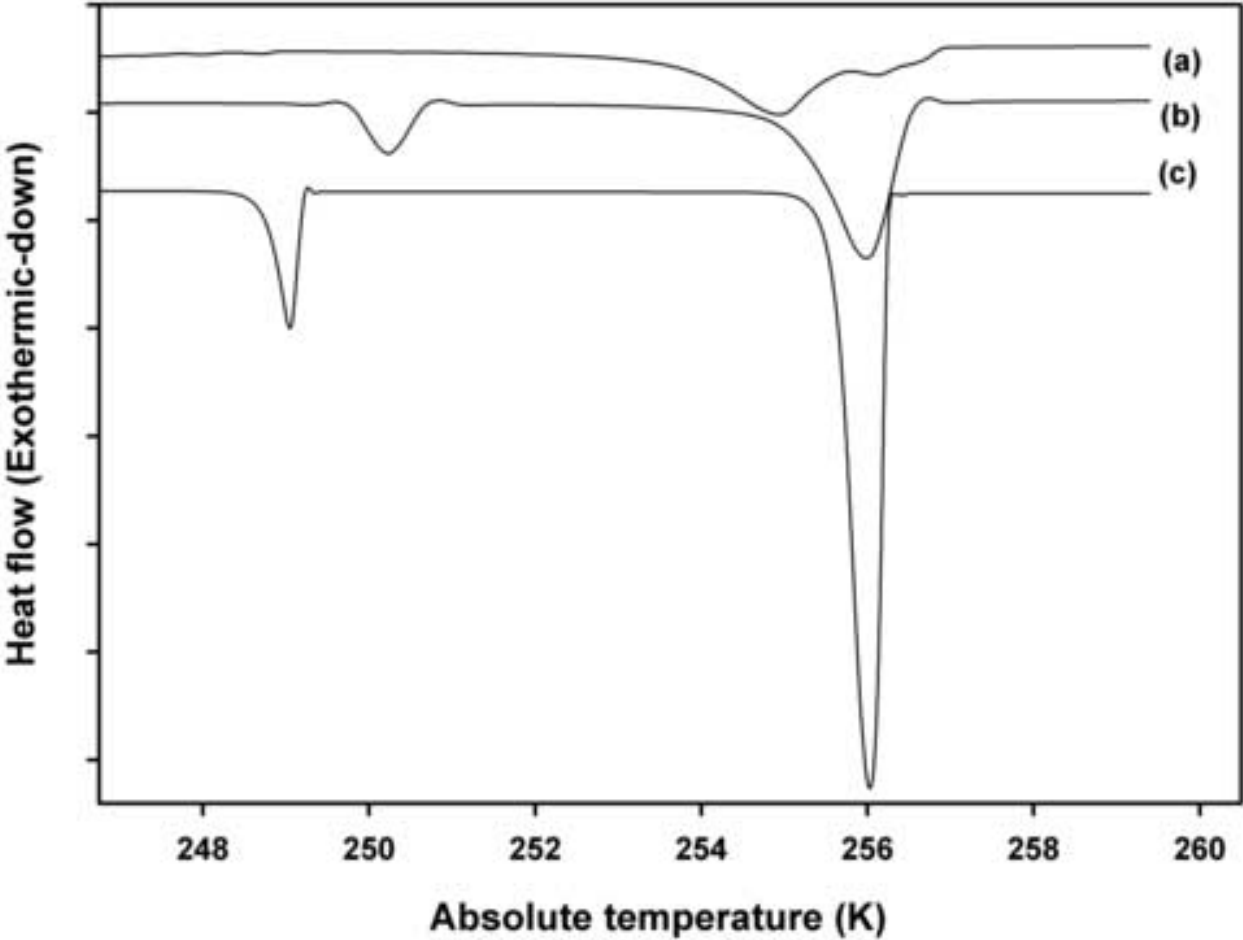


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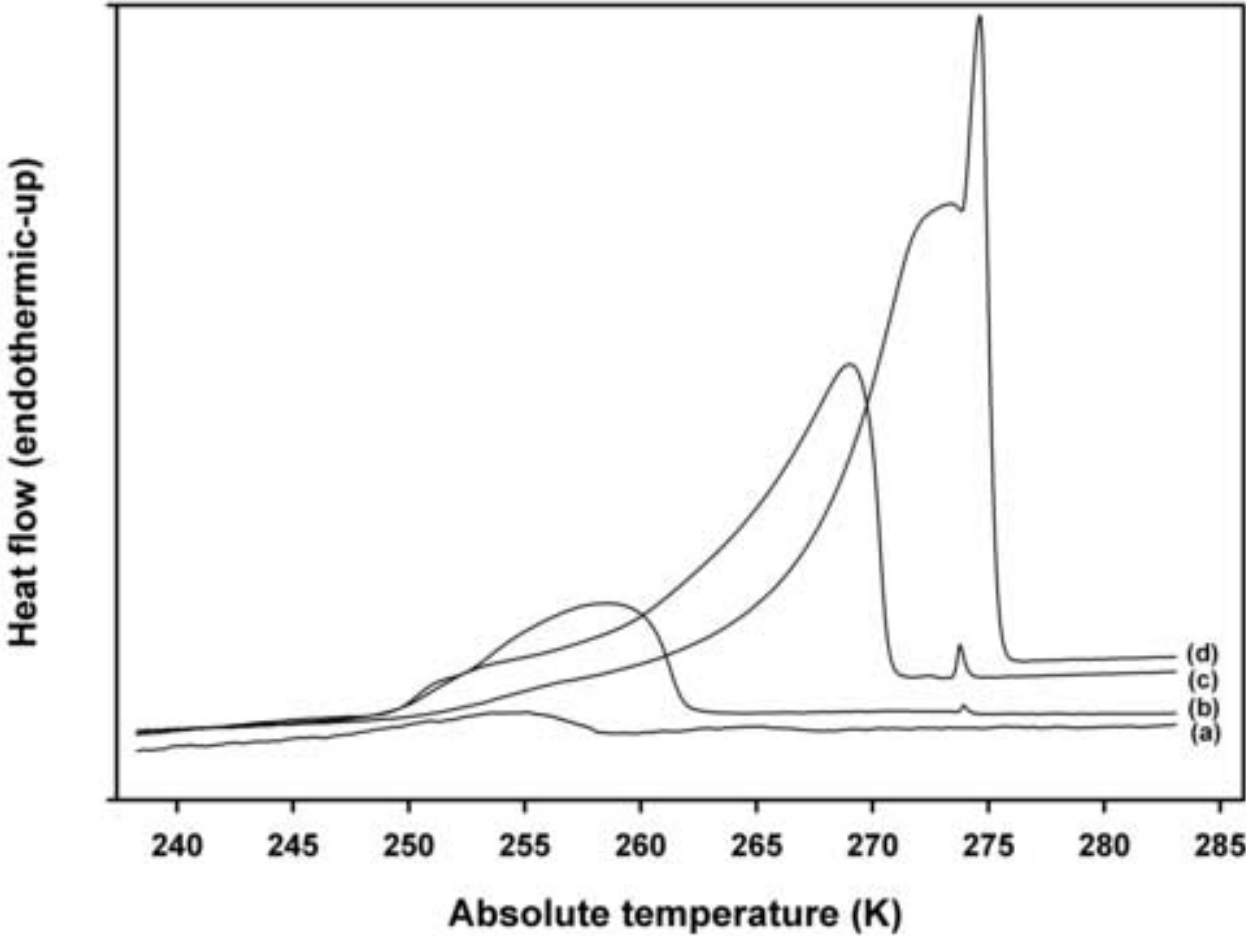


Figure 5  
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